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The Effects of Screen Sizes on the Surface Properties of Tepered Steel Treated by Active Screen Plasma Nitriding

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Abstract. Active screen plasma nitriding (ASPN) is a novel thermochemical surface treatment which has many specialities compared to the conventional direct current plasma nitriding (DCPN). The edge effect is eliminated, the hardness and layer thickness on the surface are homogeneously distributed. In this study, 42CrMo4 alloyed steel was treated with different screen sizes (screen diameter, hole size) active screen plasma nitriding. The parameters of treatment are similar in all cases (4 hours at 490°C and 2,2 torr). The nitrided samples were characterized by atomic force microscope (AFM) to analyse the roughness of the surface. The cross-section hardness of the samples and the thickness of the nitrided layers were also measured.

1. Introduction

The nitriding process is widely used for improving the hardness and the wear resistance of steels. DCPN has become a standard industrial technology, but it has some technological issues such as edge effect and hollow cathode effect [1–3]. Over the years, ASPN appeared in the plasma nitriding treatment processes. The main advantages are based on the replacement of the glow discharge region. This means, that the plasma forms on the screen, instead of the workpiece, which causes homogeneous hardness and layer thickness on the surface of the specimen, therefore the issues of the DCPN are eliminated. Considering that the plasma is formed only on the screen, and the worktable is isolated from the voltage source, the samples are heated by radiation. ASPN also generates nitrogen mass transfer to the surfaces of the specimens [4, 5]. Several researchers have reported the effects of the nitriding parameters, such as the distance between the screen and the sample [6–8], the effect of different screen sizes [9–11] on the surface roughness [12–16] and also wear and corrosion resistance improvement [17–20].

Surface roughness parameters, which can be quantified with AFM, are frequently used as a typical measurement of mechanical surface properties. In our particular case, the interaction between ions and the sample surface during the sputtering corresponds to the variation of the roughness [21–23]. The wear and corrosion resistance are depending on the base material, but these material properties are not evaluated in this research.



The aim of this research is to investigate the influence of the active screen hole sizes on the layer thickness, cross-section hardness profile and most importantly the surface roughness of the active screen plasma nitrided samples.

2. Materials and methods

In this study, tempered 42CrMo4 low alloy steel was used with the following chemical compositions shown in Table 1. The sample disks were 20 mm in diameter and 6 mm in thickness. The surface of the samples was mechanically ground with 80 to 2500-grit SiC paper and polished with 3 μm diamond suspension. The samples were cleaned with acetone and dried before placed in the vacuum chamber.

Table 1. Chemical composition of materials in wt.%

	C	Si	Mn	P	S	Cr	Mo
42CrMo4	0.38 – 0.45	≤ 0.4	0.6 – 0.9	≤ 0.025	≤ 0.035	0.9 – 1.2	1.5 – 0.3

The active screen was made of 1.0330 type steel. The screen dimensions were $\varnothing 100 \times 85$ mm with $\varnothing 5$ mm, $\varnothing 18$ mm and $\varnothing 45$ mm holes and the wall thickness was 0.8 mm. Figure 1 shows the schematic workspace of the chamber and the screens with the different hole diameters and the distance of their centers. All treatments were carried out using the same parameters as shown in Table 2 and following the same procedures. First, the chamber was pumped to a base pressure of 2×10^{-1} torr, then the chamber was flushed with argon. After the flushing, the pressure was set to 2,2 torr with the nitriding gas. The plasma was produced on a negatively polarized screen (cathode) and the anode (base of the chamber), which was held at ground potential. The temperature was monitored using an isolated K-type thermocouple under the workpiece. After the nitriding process, the workpiece was cooled down from the treatment temperature to the room temperature.

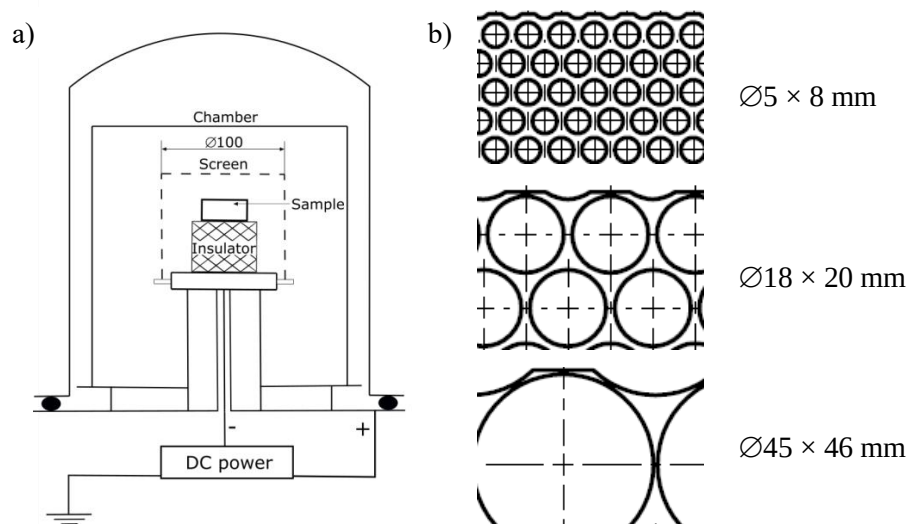


Figure 1. a) Schematic diagram of the ASPN chamber
b) Hole diameters of active screens

Table 2. Parameters of plasma nitriding

Voltage (V)	Current (A)	Gas mixture N ₂ :H ₂ (%)	Temperature (°C)	Pressure (torr)	Time (h)
490 – 540	0.9 – 1.5	25:75	490	2.2	4

Scanning electron microscope (SEM) was used to examine the morphology of the samples and measure the thickness of the nitrided layer. For the microhardness measurement Buehler IndentaMet 1105 equipment was used with Vickers head. The measuring load was 10 g for 11 s time. Contact mode AFM images were taken with a Veeco (lately Bruker) diInnova scanning probe microscope (SPM) with Bruker DNP-10 probes, using the longest cantilever with the smallest spring constant ($k = 0.03$ N/m). The sampling rate of the image acquisition was 512×512 with 1 Hz scan rate. The obtained images were post-processed with the Gwyddion 2.36 software [24]. Only standard background correction was applied on the images to remove piezo movement and sample tilt effects.

3. Results and discussion

The results of the different hole sized ASPN processes are compared with DCPN on the same nitriding parameters.

3.1. Thickness and hardness measurements

Scanning electron microscope was used to determine the depth of the compound zone. Figure 2 shows the cross-sections of the samples. The layer thickness was $5.8 \mu\text{m}$ with DCPN, $3.2 \mu\text{m}$ with $\varnothing 5$ mm holes of AS (active screen), $5.5 \mu\text{m}$ with $\varnothing 18$ mm holes of AS and $5. \mu\text{m}$ with $\varnothing 45$ mm holes of AS. The layer thickness doesn't increase significantly when the hole size is increased over $\varnothing 18$ mm. It means that there is a critical transition hole size between $\varnothing 5$ mm and $\varnothing 18$ mm which needs further investigation. The highest thickness is at DCPN treatment because of the direct heating with current. However, in this case the layer is not uneven, and nitride networks were founded at some place of the diffusion zone along the grain boundaries.

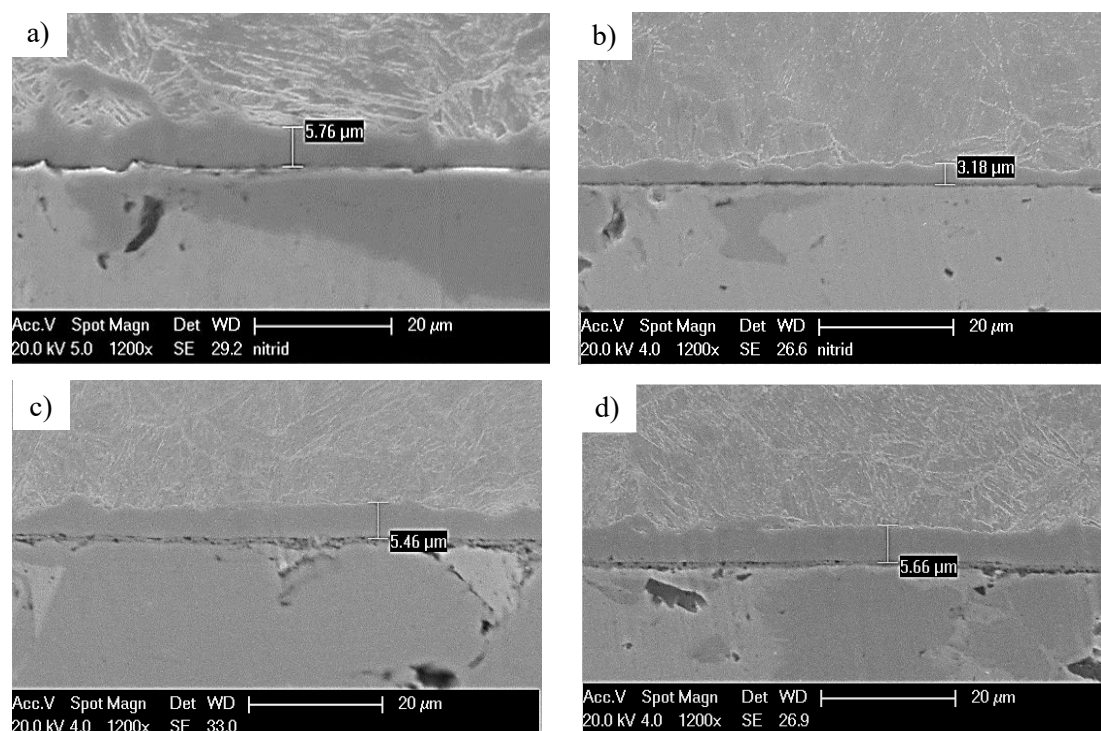


Figure 2. Compound layer thickness of steel 42CrMo4 a) DCPN, b) $\varnothing 5$ mm holes of AS, c) $\varnothing 18$ mm holes of AS, d) $\varnothing 45$ mm holes of AS treated samples

Figure 3 shows the cross-section hardness profiles of the investigated samples and the maximum hardnesses are found in Table 3. Each curve is made from 3 different measurements on different places

of the cross-section, but all started from the layer, through the diffusion zone, towards the base material. The standard deviation is also shown on the diagram in every point.

Table 3. Maximum hardnesses of nitrided samples

	DCPN	Ø5 mm holes of ASPN	Ø18 mm holes of ASPN	Ø45 mm holes of ASPN
HV0.01	1050	787	912	698

From the diagram it can be read that according to the layer thickness values, the DCPN sample has the highest hardness. This seems good, but the DCPN method always creates an edge effect, where the layer is not uniform with the other parts of the sample and the hardness values near the edge effect are differ from the common hardness of the nitrided sample [6, 8, 20]. It is also interesting, that the Ø18 mm hole size screen effects the highest hardness values among the ASPN treated samples. It can be declared as an optimum among the other screens. This statement is also backed by the layer thickness measurements. The layer thickness depends on the quantity of formed nitrides. Nitrides have significantly higher hardness than the raw material, so theoretically the thicker layers should have higher hardness. However, the hardness values of the Ø45 mm hole size ASPN treated sample should be considered. A possible explanation for the lower values, is that the holes are too large in the screen, so it is not able to create a fully even layer where there is and open area of the screen. Although this possibility should be examined further.

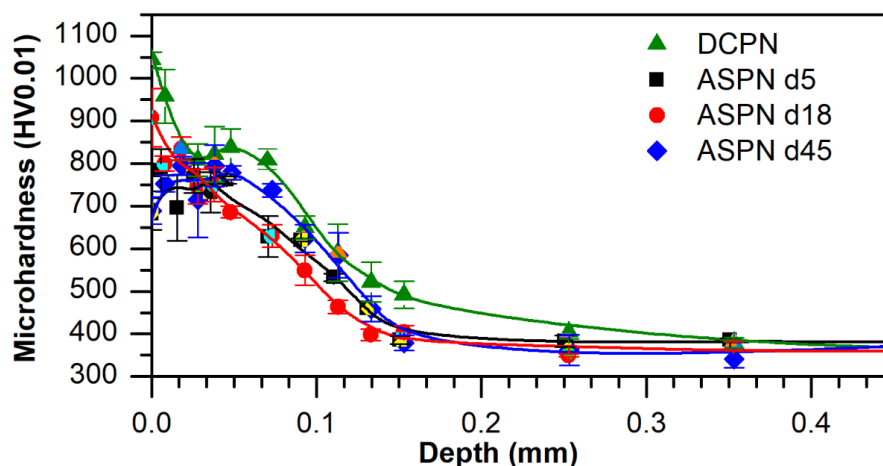


Figure 3. Hardness profile of plasma nitrided samples

3.2. Surface roughness measurements with AFM

The surface topography of the samples was investigated by AFM, as shown in figure 4. It can be clearly seen that the surface roughness of the nitrided samples can even be 30-40 times higher than the polished reference sample (S_a is 0.75 ± 0.05 nm for the reference, 15 ± 0.05 nm for the DCPN sample and is between 35-40 nm for the ASPN samples, measured on 2×2 μm images). The Ø18 mm hole size screen has the highest roughness. If the dimension of the hole size is increasing, the roughness is also increasing until the critical transition size. Each results of previous measurements confirm the Ø18 mm hole size screen is a transition size among the hole size dimensions. It has to be noted that the surface roughness is not sufficient alone to conveniently characterize the effect of nitriding. As can be seen in figure 4, the surface after the treatment is structured in different levels. Besides the grains caused by the process, we can observe surface waviness with hills and valleys (especially for the DCPN treated samples in figure 4 c)-d)), and also larger, complex structures (which resemble aggregated grains, most visible in the 10×10 μm images of figure 4 e)-j) for the ASPN samples with larger hole diameters). This means, that the treated surfaces have different characteristic features depending on the technological parameters,

and since these features have different spatial frequency, based on the selected scan-size, the measured surface roughness values will be affected differently. By using smaller scan sizes it is possible to focus on the grainy nitride structures. It is clear that the ASPN samples have significantly larger grain sizes compared to the DCPN sample, moreover, the treatment with $\varnothing 18$ mm hole size resulted in the largest average grains. At larger scan sizes the mentioned complex structures define the surface roughness, which increases with a factor of 1.4-2.3 by increasing the scan size to $10 \times 10 \mu\text{m}$ from $2 \times 2 \mu\text{m}$, depending on the samples. These structures with smaller spatial frequency could also be attributed to aspects of the treatment, but their detailed characterization is out of the scope of the current paper.

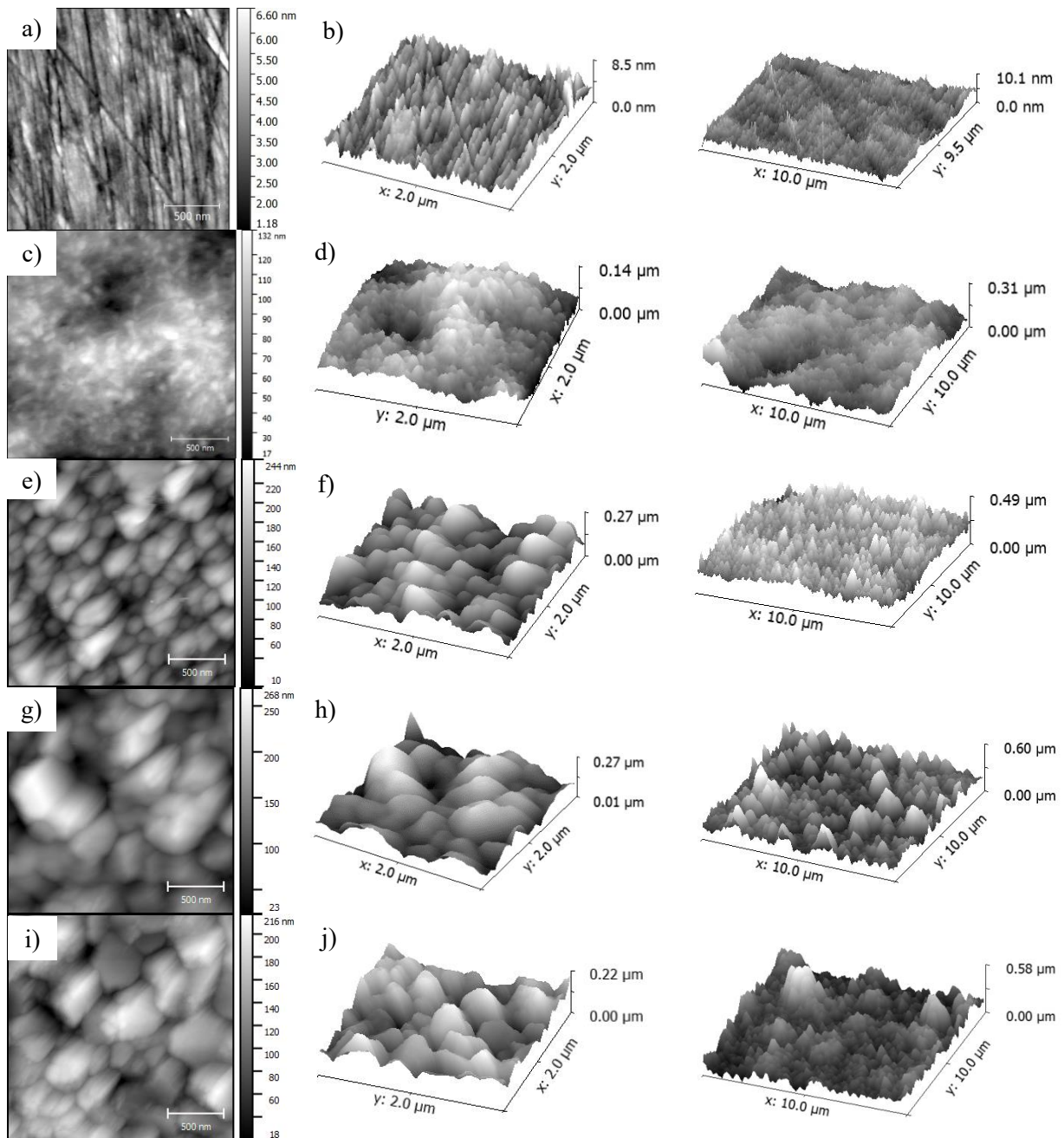


Figure 4. Contact-mode 2D and 3D AFM topography images made on a)-b) polished, c)-d) DCPN treated, e)-f) $\varnothing 5$ mm holes of AS treated, g)-h) $\varnothing 18$ mm holes of AS treated, i)-j) $\varnothing 45$ mm holes of AS treated samples.

4. Conclusions

- The highest thickness is obtained by using DCPN treatment, but the edge effect causes an uneven layer thickness. It seems that the ASPN process has a critical transition from the point of view of the screens hole sizes, which treatment influences the layer thickness.
- Also, the DCPN treatment resulted the highest hardness in the nitrided layer. The ASPN treated samples have similar hardness, but their maximum hardnesses are lower than the DCPN treated sample. After all, the highest hardness formed when the screen with Ø18 mm holes was used.
- The surface roughness of the nitrided samples can even be 30-40 times higher than the polished originals, depending on the process, but besides the nitride grains the ASPN samples show characteristic features with smaller spatial frequency. The relation of these structures with the technological parameters should be investigated in more detail in the future.
- The AFM investigation of the surfaces showed that the sample treated by using an Ø18 mm hole size screen has the highest average grain size.
- All things considered, Ø18 mm hole could be considered as the optimal size of the screen for the ASPN process, based on the investigated samples in this research.

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